

1 **Molybdenum sulphides on carbon supports as electrocatalysts for**
2 **hydrogen evolution in acidic industrial wastewater**

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18

19 **Abstract**

20

21 Microbial electrolysis cells (MECs) are an attractive future alternative technology to generate
22 renewable hydrogen and simultaneously treat wastewaters. The thermodynamics of hydrogen
23 evolution in MEC can be greatly improved by operating the cathode at acidic pH in combination
24 with a neutral pH microbial anode. This can easily be achieved with acidic industrial wastewaters
25 that have to be neutralised before discharge. For the hydrogen evolution reaction (HER) in acidic
26 wastewater, efficient and inexpensive catalysts are required that are compatible with the often
27 complex chemical composition of wastewaters. In this study, molybdenum sulphides (MoS_x) on
28 different carbon supports were successfully used for hydrogen evolution in different acidic media.
29 At first, the cathodes were screened by linear sweep voltammetry in sulphuric acid (pH 0) or
30 phosphate buffer (pH 2.2). After this, the overpotentials for H_2 production of the best cathodes and
31 their long term performances (≥ 1 week) were determined in acidic industrial wastewater (pH 2.4)
32 obtained from a plant mainly producing cellulose acetate. For the most promising MoS_x cathodes,
33 the overpotentials for HER (at $3 \text{ mA}\cdot\text{cm}^{-2}$) were only ~ 40 mV higher than for a platinum electrode.
34 Most importantly, the catalytic efficiency of the MoS_x electrodes improved in the wastewater over
35 time (7-17 days), while Pt electrodes were found to be slowly deactivated. Thus, MoS_x emerges as
36 an affordable, efficient and especially durable electrocatalyst for HER in real acidic wastewaters
37 that could take energy production from wastewaters in the form of hydrogen towards practical
38 applications.

39

40 **Keywords:** hydrogen, industrial wastewater, microbial electrolysis cell, molybdenum sulphide

1. Introduction

Hydrogen, produced in a renewable way, is considered as a potential future energy carrier. While the majority of today's hydrogen is still generated from fossil fuels by steam reforming, water electrolysis using renewable electricity (e.g. solar or wind power) is regarded as one of the key technologies for the sustainable production of hydrogen [1]. The energy required for hydrogen production through water electrolysis can be significantly lowered by choosing an alternative anode reaction that is thermodynamically more favorable than the oxygen evolution reaction. One such synergistic reaction is the bioelectrochemical oxidation of organic carbon (e.g. acetate) present in wastewaters by means of a microbial anode. Such systems with bacterial catalysts at the anode are called microbial electrolysis cells (MECs) and are currently being developed as an efficient technology to simultaneously treat wastewaters and produce hydrogen. In MECs, the biological oxidation of organic components at the anode (usually at neutral pH) produces the reducing equivalents used for H₂ evolution at the cathode, where the pH can range from 2 to 12.5 [2]. At neutral pH, the theoretical voltage difference required to drive the cathodic H₂ evolution reaction (HER) in combination with a microbial anode for acetate oxidation is only 0.11 V. This value is more than ten times lower compared to the theoretical minimum of 1.23 V required for the conventional electrolysis of water, where H₂ production at the cathode is coupled to anodic oxygen evolution. Furthermore, if an MEC is operated with a pH gradient between anode and cathode, the theoretically required voltage can be reduced by 59 mV for each pH unit that the acidic cathode deviates from neutral pH required at the microbial anode. While in theory a difference of two pH units between an acetate-oxidising anode and a hydrogen evolution cathode would be sufficient to drive the electrode reactions with zero voltage, irreversible losses occur and larger voltages are required partly due to overpotentials (η) at the cathode side.

64 Hydrogen production in an MEC with a pH difference between anode (pH 7) and cathode (pH 2)
65 has previously been reported by Ruiz et al. [2]. However, their study was conducted with
66 “laboratory electrolytes” that are not relevant to practical applications, where wastewater serves as
67 electrolyte both at the anode and cathode chamber. The type and characteristics of the wastewater
68 dictates the operational conditions at the cathode, and thus the overpotential of HER. It is therefore
69 essential to develop efficient, durable, and affordable catalysts that decrease the overpotential of the
70 HER under the conditions for practical relevance. The high cost of efficient cathode materials for
71 HER is regarded as one of the main limiting factor in scaling-up of MECs [3].

72
73 In conventional water electrolysis cells, noble metals such as platinum are often used as HER
74 catalysts. However, the excellent catalytic properties of platinum (η 's close to 0 mV at some
75 conditions [4,5]) come with a high cost of currently $\sim 6800 \text{ US\$} \cdot (\text{mol Pt})^{-1}$ (<http://www.lme.com>,
76 accessed on 22.07.2016). In addition, the poisoning of Pt-based HER catalysts in MECs has been
77 reported and might be caused by sulphide [6] or due to the adsorption of phosphate on the electrode
78 surfaces [7]. In MECs, some alternative HER catalyst materials have been tested with real
79 wastewaters as electrolytes, including stainless steel [8,9] and nickel [10,11]. However, stainless
80 steel electrodes have high overpotentials and can also encounter poisoning [12] or corrosion [13].
81 Nickel catalysts have shown promising results for H_2 evolution at near neutral pH [14,15] but are
82 not suitable for acidic pH values. There are no studies addressing HER in real wastewaters with
83 acidic pH values.

84
85 Molybdenum sulphides (MoS_x) have recently gained attention due to their high catalytic activity for
86 HER, their excellent chemical stability [16–18] and the very low cost of $\sim 1.5 \text{ US\$} \cdot \text{mol}_{\text{Mo}}^{-1}$

87 (<http://www.lme.com>, accessed on 22.07.2016). Molybdenum sulphides can be generally described
88 as ionic compounds containing Mo^{4+} cations and sulphide (S^{2-}) as well as disulphide (S_2^{2-}) anions. A
89 simplified general formula $\text{Mo}^{\text{IV}}(\text{S}^{2-})_a(\text{S}_2^{2-})_b$ can be formulated for these materials, for example, with
90 $a = b = 1$ in the case of amorphous MoS_3 , i.e. $\text{Mo}^{\text{IV}}(\text{S}^{2-})(\text{S}_2^{2-})$ [19]. Additionally, the structural motif
91 of Mo cations embedded in a sulphide-rich coordination sphere can also be found in the active site
92 of the FeMo-nitrogenase enzyme, where the FeMoCo active site produces H_2 as a side-reaction
93 while converting N_2 to NH_3 (for a review, see Laursen et al. [20]). For molybdenum sulphides, it
94 has been shown that hydrogen atoms bind easily to sulphur anions at the edges of MoS_x -layers,
95 where the HER reaction is thought to take place [5]. In MECs, electrodes with MoS_2 catalyst have
96 been reported to have H_2 production rates comparable to Pt electrodes at neutral pH in synthetic
97 medium, thus making molybdenum sulphide a promising catalysts for H_2 evolution [21].

98
99 However, MoS_x has two major limitations: 1) a low electric conductivity of the material and 2) a
100 rather low number of active sites for HER catalysis [20–22]. To overcome these drawbacks, small
101 MoS_x particles are often deposited on electrically conductive support materials by electrochemical
102 deposition [23,24] or with the help of Nafion [25,26]. For MECs, MoS_x electrodes have been mainly
103 prepared by mixing MoS_2 with Nafion and carbon black that is used to coat carbon cloth [12,27].
104 Nafion is, however, expensive, which might prevent its use in practical applications. Regarding the
105 scaling-up of MECs, carbon based support materials such as carbon cloth are particularly attractive
106 due to their low cost, sufficient electrical conductivity, and large specific surface area. Xia et al. [28]
107 reported that the conductivity and current production capability of MoS_2 catalysts were affected by
108 the concentration of MoS_2 on the electrode. In this way, the number of active sites and the
109 material's conductivity could be improved, e.g., by increasing the number of electrodeposition

110 cycles [29] or by using suitable support materials, such as carbon nanotubes (CNTs) [30]. Thus, the
111 electrode preparation process appears to be a very important step as it affects the composition,
112 particle morphology and distribution of the MoS_x deposited on the electrode surface. In addition, it
113 influences the costs of the manufactured MoS_x electrodes. Thus, novel ways to prepare MoS_x
114 electrodes are required. MoS_x electrodes have already been widely studied for possible applications
115 in conventional water electrolysis [5,28,30]. However, the requirements for MEC cathodes to
116 operate in real wastewaters are different due to generally lower conductivities of the electrolytes and
117 pH values deviating from the extremes commonly used in alkaline or acidic electrolyzers.
118 This difference in operation is compounded by the presence of various organic and sulphur-
119 containing compounds that can compete for the electrons at the cathode or cause electrode
120 poisoning.

121

122 The purpose of this study was to evaluate the use of MoS_x -based HER cathodes under application-
123 relevant conditions in an acidic industrial wastewater. The major research question addressed is
124 whether MoS_x -cathodes can sustain their high catalytic activity towards HER also in real acidic
125 industrial wastewaters over extended periods of time. Furthermore, we investigated whether the H_2
126 production potentials under these conditions are affected by the electrode preparation method, since
127 this will influence both cost and usability. Thereto different preparation methods for HER-cathodes
128 composed of MoS_x as electrocatalyst and carbon materials as conducting supports were evaluated
129 and their potential for HER were determined. Electrochemical hydrogen evolution by these MoS_x
130 electrodes was first studied in sulphuric acid and phosphate buffer to screen for the most efficient
131 cathodes and to compare their performances with earlier studies carried out under similar
132 conditions. The best HER-cathodes from this screening were then investigated in real acidic

133 industrial wastewater obtained from a plant producing to a large part cellulose acetate. This
134 cellulose derivative is produced worldwide on a large scale (especially for filter applications) by the
135 reaction of acetic acid with cellulose, often in the presence of sulphuric acid acting as acetylation
136 catalyst [31]. As a result, large amounts of acidic waste streams rich in acetic acid are available,
137 which have to be neutralised at the industrial site before they can be discharged into a sewage
138 system or wastewater treatment plant. Thus, near neutral wastewater fractions, rich in acetic acid is
139 available for the anodic oxidation in a MEC, while the acidic wastewater could be used directly as
140 electrolyte at the cathode to provide more thermodynamically favorable conditions for H₂
141 generation. To study the effects of the wastewater electrolyte on cathodic H₂ evolution, the cathodes
142 prepared in this study were tested in half-cell experiments with abiotic anodes.

2. Materials and Methods

2.1 Chemicals and supporting carbon electrodes

Molybdenum sulphide (MoS_2 powder, $< 2 \mu\text{m}$) and ammonium tetrathio-molybdate ($(\text{NH}_4)_2[\text{MoS}_4]$) were purchased from Sigma Aldrich (Germany). Different carbon based electrodes, i.e. activated carbon cloth (ACC), buckypaper (BP) and electrospun carbon (ES), were used as support materials for MoS_x deposition. Activated carbon cloth (C-TEX 13) was purchased from MAST Carbon (Basingstoke, UK) and was used without pretreatment. For electrospinning, polyacrylonitrile powder (PAN) was dissolved in N,N-dimethylacetamide (DMAc, Sigma-Aldrich, Taufkirchen) by stirring at 70°C overnight. The electrospun materials were fabricated using a modified KatoTech (Kyoto, Japan) NEU electrospinning device (Fig. A.1). The modification allows hot-air assisted electrospinning as well as uniform fiber deposition on the collector. The material was spun from 13 wt% PAN ($150,000 \text{ g}\cdot\text{mol}^{-1}$, Sigma-Aldrich) by applying a voltage of 16 kV, a polymer feed rate of 0.864 mL h^{-1} , a collector rotation of 2.5 m min^{-1} , and an airflow of 5 SCFH at 40°C . The electrospun PAN fiber mats were carbonized by heating first to 230°C in air and subsequently to 1100°C in N_2 atmosphere using a tube furnace (Gero, Germany).

Buckypaper electrodes were fabricated from multi-walled carbon nanotubes (MWCNTs) as described by Hussein et al. [32]. In short, 550 mg MWCNTs (Baytube C150 HP, Bayer Material Science AG, Germany) were dispersed in 1.1 L of a 1% aqueous solution of the surfactant Triton X-100 (Carl Roth, Germany), mixed for 15 min at 200-300 rpm and sonicated for 3 hours. The agglomerates of MWCNTs were removed from the solution by centrifugation at 2000 g for 15 min

166 and 135 mL of the MWCNT solution was vacuum-filtered on a nylon filter membrane (0.45 μm ,
167 Whatman, Maidstone, UK). The excess Triton was removed from the MWCNT film by several
168 washing steps with water, isopropanol and acetone (for further information, see supplementary
169 material of Kipf et al. [33]). After two days of washing (six washing steps) the buckypapers were
170 dried at 70°C overnight and weighed (MWCNT loading of $1.2 \pm 0.2 \text{ mg cm}^{-2}$). For comparison,
171 platinum/carbon electrodes ($0.5 \text{ mg Pt cm}^{-2}$, i.e. $2.6 \mu\text{mol cm}^{-2}$, Freudenberg H2315 I2C6, Quintech,
172 Germany) and stainless steel electrodes (SS, SIKA-R1 AX, GKN, Germany) were used. Before the
173 experiments, stainless steel electrodes were cleaned by sonication for 10 min each in acetone,
174 isopropanol, and then twice in distilled water.

175

176 **2.2 Electrode preparations**

177

178 For a list of the different MoS_x -electrodes prepared for this study, see Table 1. Drop coating on
179 supporting materials was carried out using a solution containing 1.5 mg MoS_2 in 1 mL ethanol and
180 sonicated for 10 min. 130 μL of this suspension was drop-coated on the support material (1 cm^2)
181 and heated at 80°C overnight. The resulting loading was calculated to be $0.2 \text{ mg MoS}_2 \text{ cm}^{-2}$
182 ($2.1 \mu\text{mol cm}^{-2}$). In some experiments, ES and BP electrodes were acid treated with HCl (pH 5) for
183 30 min before drop-coating. Impregnation (drop coating) on acid treated supporting materials was
184 done as described above with $(\text{NH}_4)_2\text{MoS}_4$ (concentration of MoS_2 in ethanol was 1.5 mg mL^{-1})
185 instead of MoS_2 followed by heat treatment. The resulting loading was calculated to be 0.2 mg
186 $\text{MoS}_2 \text{ cm}^{-2}$ ($2.1 \mu\text{mol cm}^{-2}$). Electrodeposition on the three supporting materials was done with a
187 method modified from Zhang et al. [34]. For this, the electrode was first soaked in electrolyte
188 ($0.05 \text{ M } (\text{NH}_4)_2\text{MoS}_4$ in 0.1 M KCl) for 15 min and simultaneously made anoxic by purging the

189 solution with N₂. The current density was then set to -8.3 mA cm⁻² for 15 min to reductively deposit
 190 Mo^{IV}S_x from the soluble Mo^{VI} precursor. Fig. A.2 shows a typical cathode potential response.

191

192 **Table 1** Electrodes investigated in this study.

Electrode substrate (sample abbreviation)	MoS_x precursor	Preparation method
Activated carbon cloth (ACC)	-	-
Activated carbon cloth (ACC-ED)	(NH ₄) ₂ MoS ₄	Electrodeposition
Activated carbon cloth (ACC-DC)	MoS ₂	Drop coating
Buckypaper (BP)	-	-
Buckypaper (BP-ED)	(NH ₄) ₂ MoS ₄	Electrodeposition
Buckypaper (BP-DC)	MoS ₂	Drop coating
Buckypaper (BP-IHT)	(NH ₄) ₂ MoS ₄	Impregnation followed by heat treatment
Buckypaper (BP-MoS ₂)	MoS ₂	MoS _x mixed with MWCNTs before buckypaper preparation
Electrospun carbon nanofibers (ES)	-	-
Electrospun carbon nanofibers (ES-ED)	(NH ₄) ₂ MoS ₄	Electrodeposition
Electrospun carbon nanofibers (ES-DC)	MoS ₂	Drop coating
Electrospun carbon nanofibers (ES-IHT)	(NH ₄) ₂ MoS ₄	Impregnation followed by heat treatment
Electrospun carbon nanofibers (ES-MoS ₂)	(NH ₄) ₂ MoS ₄	MoS _x mixed with MWCNTs before buckypaper preparation

193 MWCNT = multi-walled carbon nanotubes

194

195 Buckypapers were also prepared by mixing 100 mg MoS₂ into the MWCNT / Triton mixture before
 196 sonication. The buckypaper preparation procedure was otherwise the same as described above and
 197 should result in MoS₂ loading of 0.2 mg cm⁻² (2.1 μmol cm⁻²). Electrospun electrodes were also
 198 prepared by mixing MoS₂ (Sigma Aldrich) directly into the PAN/MDAc mixture used for the
 199 electrospinning process. The material was then spun from 9 wt% PAN (200,000 g mol⁻¹, Dolan,
 200 Germany) containing 0.24 wt% MoS₂ particles (Sigma-Aldrich, Taufkirchen, Germany) by applying
 201 a voltage of 18.5 kV, a polymer feed rate of 0.864 ml h⁻¹, and a collector rotation of 9 m min⁻¹. The

202 following carbonization step was then carried out as described above. The resulting MoS₂ loading
203 was calculated to be approximately 0.2 mg cm⁻² (2.1 μmol cm⁻²).

204

205 **2.3 Electrochemical characterization**

206

207 Linear sweep voltammetry (LSV) was used to test the activity of the modified electrodes (Table 1)
208 towards HER in three electrolytes, i.e., 0.5 M H₂SO₄, 0.5 M phosphate buffer, and acidic industrial
209 wastewater. The LSVs were used for comparing the electrodes and for the evaluation of MoS_x
210 deposition methods and not to give absolute values - therefore only one repetition per modified
211 electrode was carried out. The experiments were conducted at 30°C and for each experiment a new
212 electrode with working area of 1 cm² was used. The pH and conductivities of the electrolytes were
213 as shown in Table 2 and the effects of different electrolytes on the LSV curves are shown in Fig.

214 A.3. The acidic industrial wastewater had a chemical oxygen demand (COD) between 4.5 and 5.6 g
215 L⁻¹ (depending on the sampling time), from which 1.8-3.8 g L⁻¹ (30-63 mmol L⁻¹) was organic acids
216 (measured as acetic acid equivalents). In addition, the wastewater contained 2.0-2.8 g L⁻¹ (21-29
217 mmol L⁻¹) sulphate.

218

219 LSVs were measured in a three-electrode setup where the modified electrode was the working
220 electrode, a Pt mesh (Goodfellow, Germany) acted as counter electrode and a saturated calomel
221 electrode (SCE, +0.244 V vs. SHE, KE 11, Sensortechnik Meinsberg, Germany) as reference
222 electrode. The electrodes were connected by titanium wires and a Solatron 1470E potentiostat
223 (Ametek, Farnborough, UK) was used to record LSVs from open circuit voltage (OCV) +20 mV to

224 -1.20 V (vs. SCE), which were repeated three times for each electrode. Before starting the
225 experiments, the electrolyte was made anoxic by N₂ purging for 15 min.

226

227 **Table 2** Characteristics of the three electrolytes used for LSV experiments.

Electrolyte	pH	Conductivity (mS cm ⁻¹)	Theoretical H ₂ evolution starts at (mV vs. SCE)
0.5 M H ₂ SO ₄	0.2	200	-241
0.5 M phosphate buffer	2.2	38	-368
Industrial wastewater	2.4 ± 0.2	6.1 ± 0.6	-380

228

229 The LSV results were corrected for the ohmic potential drop (iR, uncompensated resistance), which
230 was analysed with a current interrupt method in phosphate buffer and industrial wastewater. In this
231 method, untreated electrodes (1 cm²) were placed in the different electrolytes using the three-
232 electrode set-up described above. A current of 1 mA was applied for 4 s with a potentiostat
233 (PCI4/300 Model, Gamry Systems, Pennsylvania) that enables a small sampling period of 60 μs.
234 The voltage drop upon current interruption was measured and used for calculating the ohmic
235 potential drop. The mean iR values (± standard deviation) for 0.5 M phosphate buffer and acidic
236 industrial wastewater were 89±8 Ω and 240±8 Ω, respectively. For measuring the uncompensated
237 resistance in 0.5 M H₂SO₄, the LSVs were run in the three electrode set-up by placing the reference
238 electrode with a Luggin capillary in front of the electrode or to its normal position. The ohmic
239 resistance between these two runs were measured from the linear regions of the LSV curves and the
240 difference was regarded as the ohmic potential drop. The thus obtained iR values were 0.01 Ω,
241 5.7 Ω, 9.9 Ω and 7.5 Ω for ACC, BP, ES and Pt electrodes, respectively.

242

243 In each case, the third run of the LSVs was used to compare the results. All the potentials are
244 reported against RHE, if not otherwise mentioned ($U \text{ (V vs. RHE)} = U \text{ (V vs. SCE)} + 0.244 \text{ V} +$
245 $0.058 \text{ V} \cdot \text{pH}$). Electrode performances were compared by determining the onset overpotentials by
246 drawing a tangent on the linear region of the LSV curve and reading the cathode potential at 0 mA
247 cm^{-2} . Furthermore, the overpotentials were determined at current densities of 1, 2 and 10 mA cm^{-2}
248 as described by Wirth et al. [35]. Tafel slopes were calculated for the electrodes run in $0.5 \text{ M H}_2\text{SO}_4$
249 to estimate the hydrogen evolution reaction (HER) mechanisms. The onset overpotential and
250 overpotential results were rounded to 10 mV .

251

252 **2.4 Long term experiments**

253

254 The long term behaviour of the most promising electrodes was determined by applying a constant
255 reductive current of 3 mA cm^{-2} for three parallel electrodes for at least one week or as long as the
256 cathode potential was stable (change in cathode potential $< 5 \text{ mV d}^{-1}$). The experiments were
257 conducted in a 6-electrode set-up that had a working volume of 1 L [36]. In this reactor, six half-cell
258 cathodes (1 cm^2) can be characterised simultaneously. Again, Pt meshes were used as counter
259 electrodes and SCE as reference electrode. The working and counter electrodes were connected with
260 Ti wires. As electrolyte, 0.5 M phosphate buffer or acidic industrial wastewater was used and was
261 continuously purged with humidified N_2 to ensure anoxic conditions. The current was applied with
262 an electrical testing system consisting of an electronic load (STG 2008 stimulus generator,
263 Multichannel Systems, Germany) to apply the chosen current, and a data acquisition unit (Keithley
264 2700, Keithley, Germering, Germany) to record the cathode potentials against the reference
265 electrodes [36]. The experiments were conducted at 30°C . The results were corrected for

266 uncompensated resistance as determined by numerical simulation (COMSOL) according to the used
267 reactor geometry and the position of the electrodes (reference, working and counter electrode) using
268 the conductivities of the phosphate buffer (19 mS cm^{-1}) and wastewater (6.35 mS cm^{-1}). The mean
269 iR values in phosphate buffer and industrial wastewater were $55 \pm 2 \Omega$ and $165 \pm 5 \Omega$, respectively.
270 In both electrolytes, the pH changes in long term experiments were less than 0.05 pH units over the
271 entire time of the experiments.

272

273 **2.5 Hydrogen production yields**

274

275 The hydrogen production yields of BP-ED and BP-IHT electrodes were determined in two-chamber
276 electrochemical cells with abiotic anodes as counter electrodes. The anode and cathode chambers
277 had total and working volumes of 53 and 40 mL, respectively. The compartments were separated
278 with an anion exchange membrane (fumasep FAA-PK-130, Fumatech, Germany) and reference
279 electrode (SCE, +0.244 V vs. SHE, KE 11, Sensortechnik Meinsberg, Germany) was separated from
280 the cathode compartment with a proton exchange membrane (fumapem F-950, Fumatech,
281 Germany). Each experiment was conducted with a fresh MoS_x electrode that had a working area of
282 1 cm^2 and two parallel reactors were run for each set of parameters. Pt mesh was used as counter
283 electrode at the anode. Both anode and cathode compartments contained acidic industrial
284 wastewater (Table 2) and were made anoxic by purging them with N_2 for at least 15 min prior to the
285 start of the experiments. To keep the pH stable, fresh anoxic wastewater was pumped to both
286 compartments with a flow rate of $58 \mu\text{L min}^{-1}$ resulting in a hydraulic retention time of 11.6 h. The
287 cathode potential was set to -1.2 V vs. SCE (-0.82 V vs. RHE) with a potentiostat (PCI4/300, Gamry
288 Instruments) and the current was measured. The gases produced were collected into 0.1 L gas-bags

289 (Calibrated Instruments, USA) for 1 week. The wastewater characteristics (pH, conductivity) at the
290 cathode and anode were analysed in the beginning and in the end of the experiment. The cathodic
291 hydrogen recoveries were calculated as suggested by Logan et al. [6].

292

293 **2.6 Chemical analyses and electrode characterisation**

294

295 Conductivity and pH were measured with inoLab® Multi 720 (WTW) instrument equipped with
296 TetraCon®325 conductivity probe (WTW) and SenTix®61 pH electrode (WTW). Organic acids,
297 chemical oxygen demand (COD) and sulphate in the wastewater were measured with cuvette tests
298 for organic acids (LCK365, Organic Acids Cuvette Test 50-2500 mg L⁻¹, Hach Lange, Germany),
299 COD (LCK514, Chemical Oxygen Demand test, 100-2000 mg L⁻¹, Hach Lange) and sulphate
300 (LCK153, Sulphate Cuvette Test 40-150 mg L⁻¹, Hach Lange). The gas composition in the gas bags
301 was analysed using a gas chromatograph (Clarus 480, PerkinElmer) equipped with 5 Å molecular
302 sieve column (Restek) and a thermal conductivity detector (TCD). The temperatures of oven,
303 injector and detector were 70°C, 100°C and 280°C, respectively. Argon was used as carrier gas at a
304 pressure of 290 kPa. The volume of the gas in the gas bags was analysed with a water displacement
305 method.

306

307 The electrode surfaces were imaged with a scanning electron microscope (SEM, Zeiss Supra 60 VP)
308 with a working distance of 10 mm and acceleration voltage of 5 kV. Energy-dispersive X-ray
309 (EDX) spectra were used to determine the sulphur-to-molybdenum ratios for the catalyst layers with
310 an Oxford Instruments Inca 300 EDX. For each electrode, five spots were chosen for EDX
311 measurements and an average S:Mo-ratio was calculated from the five recorded EDX spectra.

312 **3. Results and Discussion**

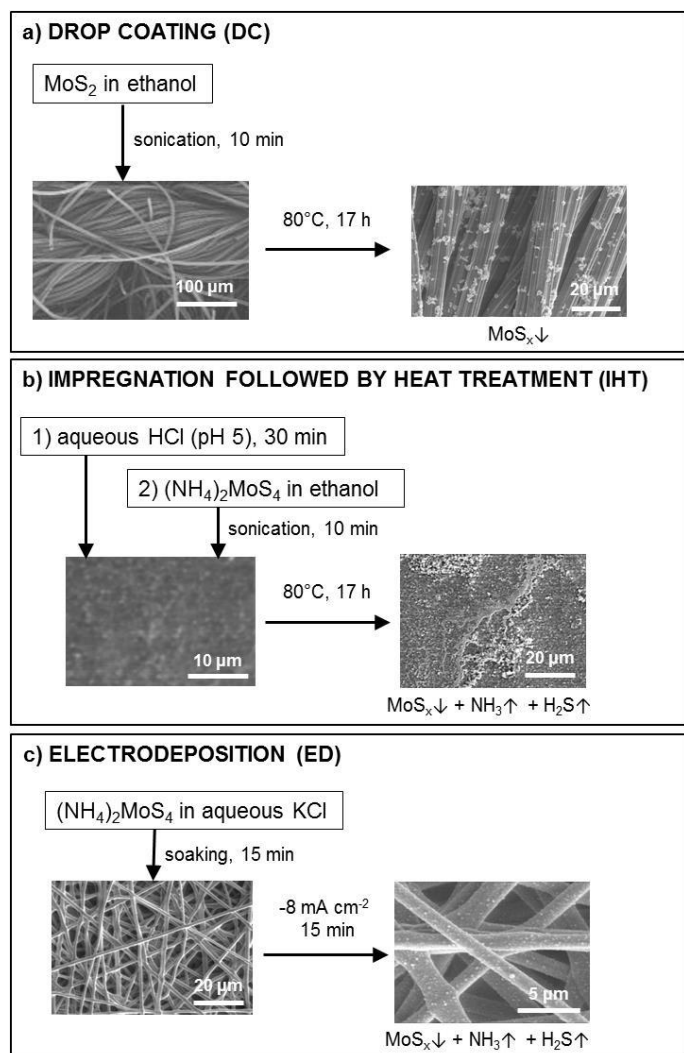
313

314 **3.1 Electrode preparation**

315

316 For the fabrication of MoS_x-coated carbon electrodes, three materials were used as carbon supports
317 for the deposition of molybdenum sulphides (MoS_x): activated carbon cloth (ACC), electrospun
318 carbon (ES) and buckypaper (BP) (Fig. 1, Table 1). Furthermore, three different deposition methods
319 that can be realised at moderate temperatures were chosen. Overall, a screening on the effects of
320 both the carbon supports and the MoS_x deposition methods was then carried out to study the
321 influence of both parameters on the electrocatalytic activity towards HER.

322



323

324 **Figure 1** Schematic illustration of deposition of MoS_x a) on activated carbon cloth (ACC) with drop
 325 coating (DC), b) on buckypaper (BP) electrode with impregnation followed by heat treatment (IHT),
 326 and c) on electrospun (ES) carbon nanofibers with electrodeposition (ED).

327

328 3.2 Electrode characterization

329

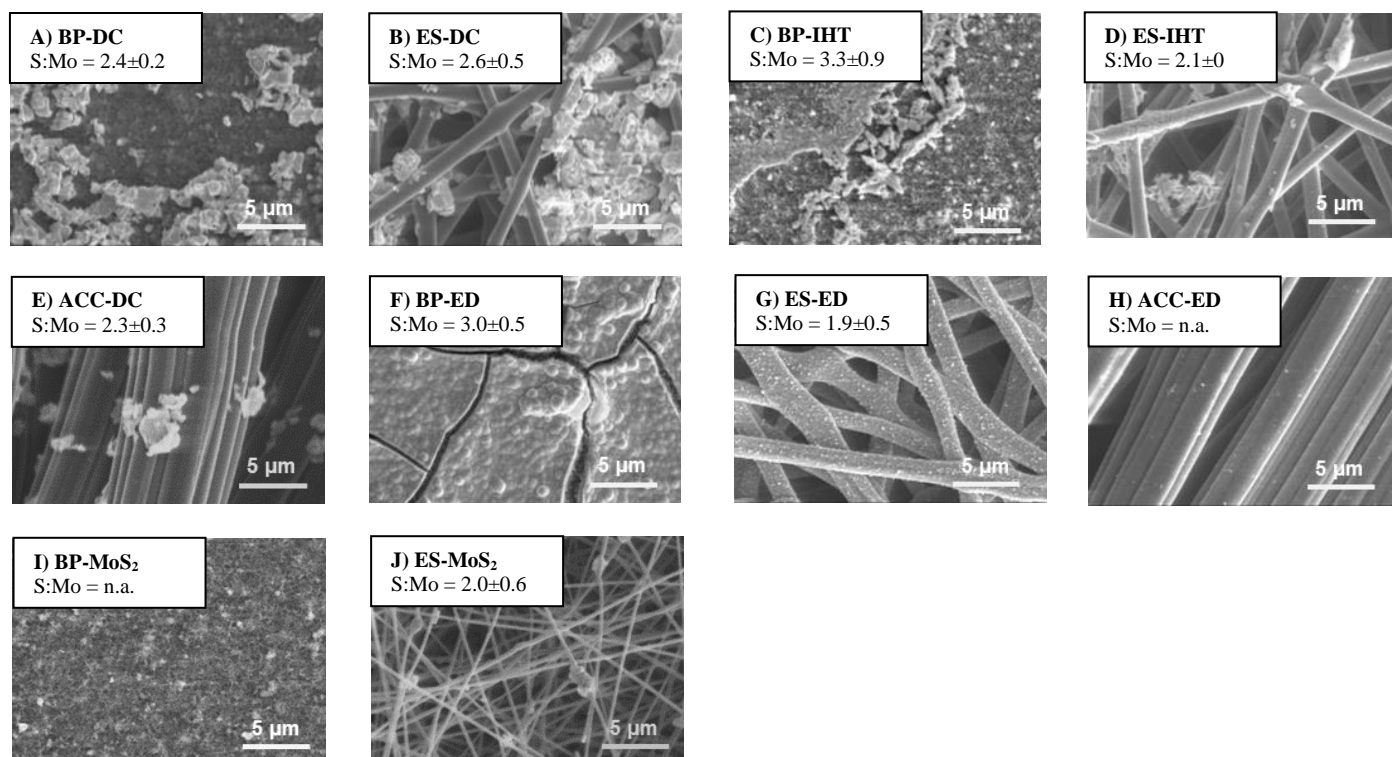
330 SEM micrographs of the electrodes after MoS_x deposition are shown in Fig. 2. In combination with
 331 scanning electron microscopy (SEM), the samples were also analysed by energy dispersive X-ray
 332 spectroscopy (EDX) to estimate the S : Mo ratio for each sample (Fig. 2). The large standard

333 deviations of the EDX results (up to ± 0.9) in these cases can be attributed either to the chemical
334 reactions on the electrode surfaces (resulting in $\text{Mo}^{\text{IV}}(\text{S}^{2-})_a(\text{S}_2^{2-})_b$ materials of variable a : b-ratios) or
335 to a large measurement uncertainty, as comparably small amounts of MoS_x were present on a large
336 background of carbon. Drop coating with MoS_2 resulted in agglomerated particles ($< 1 \mu\text{m}$) on the
337 electrode surface, while the distribution of the particles was uneven. The ratio of S to Mo in the drop
338 coated electrodes was in the range of 2.3 and 2.6 (Fig. 2.A, B, E), which is higher than the S : Mo
339 -ratio of 2.0 of the commercial MoS_2 used. Impregnation of ES and BP electrodes with solutions of
340 $(\text{NH}_4)_2\text{MoS}_4$ in ethanol followed by mild heat treatment (80°C , 17 h) resulted in smaller particles
341 covering the electrodes with S : Mo ratios of 2.6 and 3.3, respectively (Fig. 2.C,D). According to
342 Li et al. [5], heating a mixture of diluted HCl (pH5) and $(\text{NH}_4)_2\text{MoS}_4$ on CNTs at 90°C results in
343 electrostatic interactions leading to the adsorption of thiomolybdate on the CNTs. This then
344 decomposes into amorphous MoS_x at 90°C . Li et al. [5] also treated N-doped CNT (NCNT) or plain
345 CNT forest with acid followed by addition of $(\text{NH}_4)_2\text{MoS}_4$ and heating the mixture to 90°C . They
346 observed that NCNTs were coated with amorphous MoS_3 including tiny MoS_2 crystalline domains
347 that were in close contact with the NCNT surface, while the CNT surface contained infrequently
348 deposited MoS_2 patches. In our study, the electrodes seemed to contain both tiny particles and
349 patches, when carbon supports were impregnated with $(\text{NH}_4)_2\text{MoS}_4$ and heat treated (Fig. 2.C, D).

350

351

352



353 **Figure 2** SEM pictures and EDX data (ratio of sulphur to molybdenum) for electrodes prepared by
 354 different routes for depositing MoS_x on various carbon supports. For abbreviations, see Table 1.
 355 n.a.: below the EDX detection limit.

356
 357 Electrodeposition on ES and ACC electrodes resulted in an even coverage of small MoS_2 particles
 358 on the electrode material (Fig. 2.G, H). Electrodeposition on the BP electrode, on the other hand,
 359 resulted in a cracked mat built up mainly from MoS_3 that contained spherical formations (Fig. 2.F).
 360 The different molybdenum sulphide layers on the BP electrodes likely resulted from the flatter
 361 structure of the BP support (which also contains very small pores in the range of 50 nm [32]). It is
 362 likely that the same amount of MoS_2 was electrodeposited on the surfaces of BP, ES and ACC
 363 electrodes. However, as BP has a smaller accessible surface area, a layer of MoS_3 formed on top of
 364 the BP electrode instead of an even distribution of the particles. Zhang et al. [34] obtained MoS_x
 365 nanoparticles of 40-90 nm on carbon nanotube films with the same electrodeposition method.

366 However, in their study the CNTs were vertically aligned and thus provided a surface area that's
367 more accessible for MoS₂ deposition. Mixing MoS₂ to the polymer before preparation of ES
368 electrodes resulted in electrodes mainly covered by MoS₂ particles (Fig. 2.J). EDX results could not
369 be obtained for ACC-ED and BP-MoS₂ electrodes due to the low amount of Mo and S compared to
370 carbon.

371

372 Deposition of MoS_x on carbon supports resulted in S : Mo ratios varying from 1.9 to 3.3. These
373 differences likely occur due to variations of the sulphide-to-disulphide ratio in the MoS_x materials
374 (general formula Mo^{IV}(S²⁻)_a(S₂²⁻)_b). Thus, the higher the S : Mo ratio, the more sulphur is present in
375 the MoS_x in the form of S₂²⁻. Since hydrogen atoms are known to bind to the disulphide edges of the
376 MoS_x particles during catalysis [37], higher S : Mo-ratios are most likely beneficial for HER
377 catalysis. In the sample set studied here, both the MoS_x morphology and the S : Mo-ratio seem to
378 depend on the deposition method as well as the carbon support. For example, electrodeposition or
379 impregnation followed by heat treatment resulted mainly in MoS₂ deposition on activated carbon
380 cloth and electrospun electrodes, while buckypapers were coated mainly by a MoS₃ type compound
381 by these treatment methods.

382

383 **3.3 Catalytic activity towards HER in sulphuric acid**

384

385 As an initial screening, the electrocatalytic HER activity of each electrode was first determined by
386 recording linear sweep voltammograms (LSV) in 0.5 M H₂SO₄ (Fig. 3, Fig. A.4). This enabled us to
387 compare our results with literature data on the performance of MoS_x electrodes in traditional water
388 electrolysis.

389

390 The onset overpotentials for HER were determined by drawing a tangent on the linear region of the
391 LSV curve and extrapolating it towards a current density of 0 mA cm^{-2} (Fig. A.5). In addition, the
392 overpotentials were taken from the LSV data at fixed reductive current densities of 1, 2 and/or
393 10 mA cm^{-2} . As expected, the Pt electrode showed the best performance (Fig. 3) with an onset
394 overpotential of only 20 mV. The overpotentials of the Pt electrode could not be determined at low
395 reductive current densities due to capacitive behaviour of the electrode (Fig. A.5).

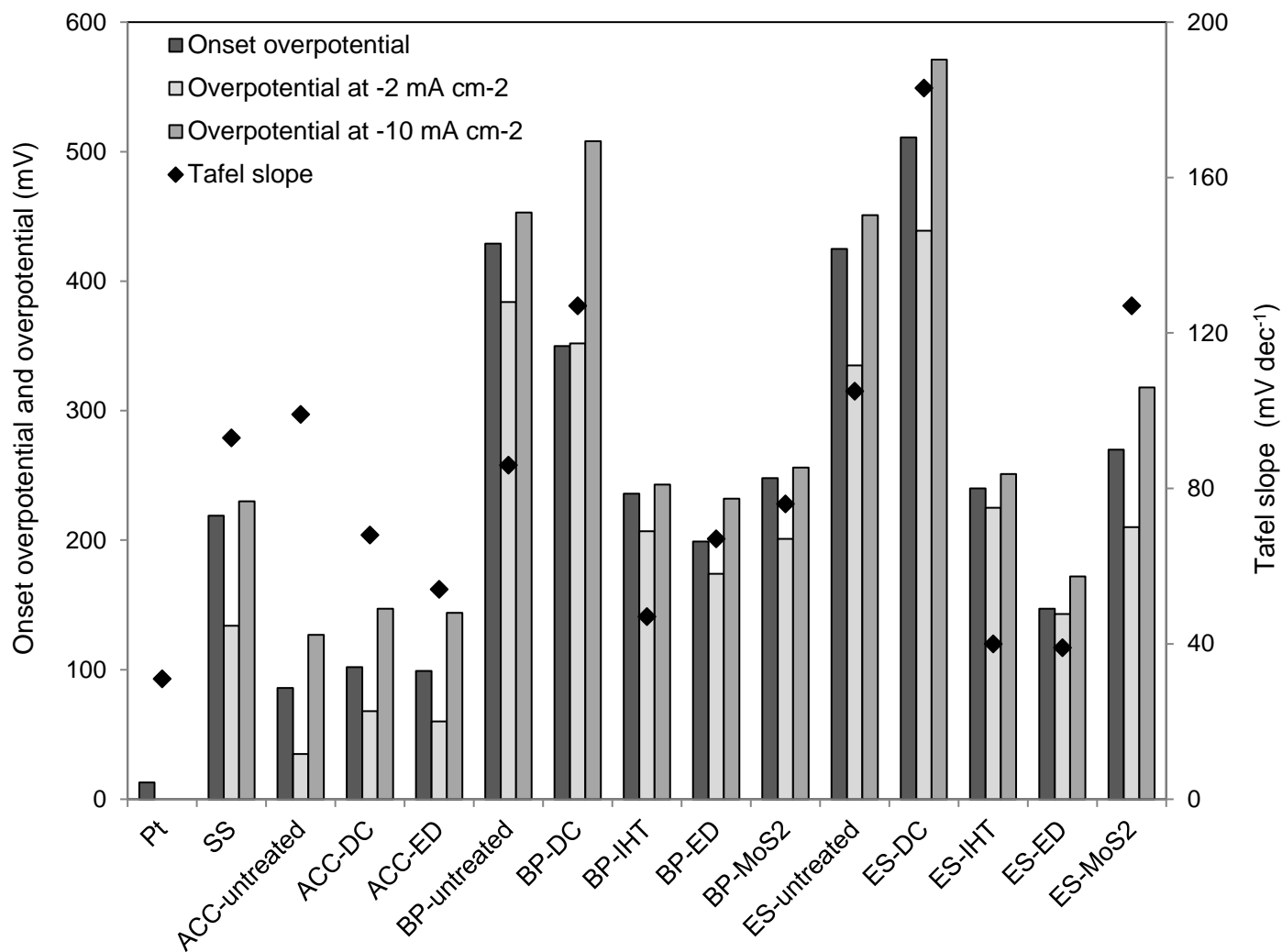
396

397 For almost all carbon electrodes the deposition of MoS_x increased the catalytic activity for H_2
398 evolution compared to untreated electrodes. The only exception was the activated carbon cloth
399 electrode which had a lower onset overpotential (20 mV) than ACC electrodes coated with MoS_x by
400 electrodeposition (40 mV) or drop coating (50 mV). The activity of the untreated ACC electrode for
401 HER was further studied by applying a constant reductive current of 3 mA cm^{-2} for triplicate
402 electrodes in $0.5 \text{ M H}_2\text{SO}_4$ for a longer time (Fig. A.6). It was observed that the overpotential of the
403 ACC electrodes increased in the first hour from negative values to around 350 mV. This suggests
404 that in the first hour some oxidised surface sites of the activated carbon cloth were reduced. This
405 could explain the initially small “overpotentials” of untreated ACC electrodes, but the detected
406 currents then actually do not represent proton but carbon reduction. After one week, the
407 overpotentials of the untreated ACC electrodes had risen to around 200 mV. Hydrogen evolution
408 with activated carbon cloths (VITO-CoRETM and VITO-CASETM), has been earlier reported in
409 phosphate buffer at pH 7 [38].

410

411 Drop coating of commercial MoS₂ on BP and ES electrodes also resulted in a lower HER activity
 412 compared to untreated electrodes. Overall, these results indicate that drop coating with MoS₂ as
 413 electrode treatment is not a suitable method to prepare HER-cathodes based on the studied carbon
 414 supports, while the other electrode treatment methods should be further studied (Fig. 3). Especially,
 415 other electrode treatments than drop coating often result in smaller MoS_x particle sizes, which
 416 increases the specific surface areas and amount of sulphur edges and results in high catalytic activity
 417 towards HER [16] compared to electrodes prepared by drop coating.

418



419

420 **Figure 3** LSV results obtained in 0.5 M H₂SO₄ (for the abbreviations see Table 1).

421

422 In sulphuric acid, electrospun carbon with electrodeposited MoS_x (ES-ED) showed the lowest HER-
423 onset overpotential of 150 mV and overpotential of 170 mV at 10 mA cm⁻² of all studied MoS_x
424 electrodes (Fig. 3). These results are lower than the values of stainless steel (220 and 230 mV,
425 respectively), a material often used as cathode in MECs. The results were further compared to
426 results reported in the literature for MoS_x electrodes in strongly acidic electrolytes (Table 3). Onset
427 overpotentials and overpotentials obtained in this study are typical for MoS_x deposited on
428 electrospun carbon, CNT and graphene electrodes. The good performance of the best electrodes in
429 this study can likely be attributed to the good conductivity and large surface area of the carbon
430 supports [28,30] as well as the good contact between MoS_x and the base material. The performance
431 of the MoS_x electrodes could be further improved by optimizing the MoS_x-loading per area [28]. As
432 MoS_x materials are typically poor conductors, the electrical accessibility of the electrode surface
433 decreases with increasing thickness of the MoS_x-layer. Thus, too high MoS_x loadings should be
434 avoided. The results indicate that the electrodes prepared for this study are comparable to MoS_x
435 electrodes prepared previously for traditional electrolysis and that our cathodes also generally show
436 a high catalytic activity towards HER.

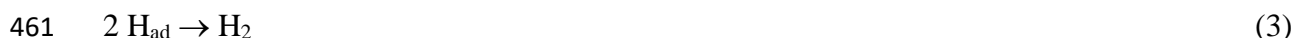
437 **Table 3** Comparison of the onset overpotentials, overpotentials and Tafel slopes obtained in this study to other MoS_x / carbon
 438 electrodes run in H₂SO₄. For the abbreviations of this study, see Table 1. The results obtained in this study were rounded to ten.

Electrode	Electrolyte	Onset overpotential (mV)	Overpotential (mV)	Tafel slope (mV dec ⁻¹)	Reference
Molybdenum carbide	0.1 M H ₂ SO ₄	333	629 at 20 mA cm ⁻²	-	[35]
MoS ₃ on GCE (CV)	pH 0	-	150 at 0.4 mA cm ⁻²	40	[29]
MoS ₂ -MWCNT on GCE	1 M H ₂ SO ₄	250	500 at 30 mA cm ⁻²	-	[28]
MoS ₂ -MC on GCE		200	500 at 16 mA cm ⁻²		
MoS ₂ -RGO on GCE		130	500 at 235 mA cm ⁻²		
MoS _x -graphene (electro-deposition on GCE)	0.5M H ₂ SO ₄	130	183 at 10 mA cm ⁻²	43	[24]
MoS ₂ on GCE	0.5 M H ₂ SO ₄	130	300 at 13.8 mA cm ⁻²	52	[39]
MoS ₃ -MWCNT drop coated on Ag electrode	1 M H ₂ SO ₄	-	150 at 1.1 mA cm ⁻² 200 at 10.8 mA cm ⁻²	40	[40]
MoS _x /NCNT	0.5 M H ₂ SO ₄	75	110 at 10 mA cm ⁻²	40	[5]
S rich-MoS ₂ -NCNF	0.5 M H ₂ SO ₄	30	120 at 10 mA cm ⁻²	38	[41]
MoS ₂ -NCNF		98	230		
MoS ₂ /CNT	0.5 M H ₂ SO ₄	90	-	44.6	[30]
MoS ₂ -RGO on GCE	0.5 M H ₂ SO ₄	100	-	41	[4]
MoS ₃ electrodeposited on GCE	1 M H ₂ SO ₄	-	170 at 20 mA cm ⁻²	-	[23]
ACC	0.5 M H ₂ SO ₄	90	130 at 10 mA cm ⁻²	100	This study
ACC-ED		100	140	50	
BP-IHT		240	240	50	
BP-ED		200	230	70	
ES-IHT		240	250	40	
ES-ED		150	170	40	

439 GCE = glassy carbon electrode, MWCNT = multi-walled carbon nanotubes, NCNF = N-doped carbon nanofibers, NCNT = N-doped
 440 carbon nanotubes, RGO = reduced graphene oxide

441 The Tafel slopes accessible from the LSV curves can be used to obtain information about the
442 HER mechanism in acidic solution (Fig. 3). The smaller the Tafel slope, the faster the increase in
443 H₂ production rate when the cathode overpotential is increased [39]. Three elementary reactions
444 are thought to be involved in H₂ evolution in acidic conditions: the Volmer (Eq. 1), Heyrovsky
445 (Eq. 2) and Tafel (Eq. 3) steps [20]. For Tafel slopes around 120 mV dec⁻¹ or higher, the Volmer
446 step is considered to be rate limiting [21]. Thus, the adsorption of hydrogen atoms on the cathode
447 surface seems to be rate limiting for untreated ACC and ES, drop coated BP and ES, and ES-
448 MoS₂. Limitations in the Volmer step might be due to a limited number of sulphur edges of the
449 MoS_x particles [16] or poor electrical contact between MoS_x and the carbon support [21]. The
450 Heyrovsky and Tafel steps proceed theoretically with Tafel slopes of 40 and 30 mV dec⁻¹,
451 respectively [21]. Thus, HER with Pt electrode proceeds via the Volmer-Tafel mechanism with
452 desorption (Tafel) as the limiting step, which is in agreement with the literature [5,24]. The
453 Volmer-Heyrovsky mechanism, on the other hand, seems to be important for the following
454 electrodes: BP-IHT, ES-IHT and ES-ED. With these electrodes, H-H bond formation and
455 desorption of the H₂ product is likely limiting the H₂ evolution. Similarly, low Tafel slopes close
456 to 40 mV dec⁻¹ have also been reported for MoS_x electrodeposited on glassy carbon electrodes
457 [24,29].

458



462

463 Based on the Tafel slopes as well as the LSV results, electrodepositions (ED) or impregnations
464 followed by heat treatment (IHT) result in the highest activity towards HER. The good
465 performance of these electrodes can be attributed to small, highly dispersed MoS_x particles on
466 the carbon electrodes ensuring strong electronic coupling of these two [4]. Drop coating did not
467 seem to be an effective way to prepare MoS_x-electrodes for HER. In addition, the Tafel slope of
468 untreated ACC electrodes was higher (100 mV) than that of other electrodes which resulted in
469 low onset overpotentials and overpotentials. This suggests that despite the low onset
470 overpotential and low overpotentials of ACC electrodes, a different mechanism for H₂
471 production is of importance here and this support material might not be the most attractive option
472 for H₂ evolution.

473

474 **3.4 Catalytic activity towards HER in phosphate buffer**

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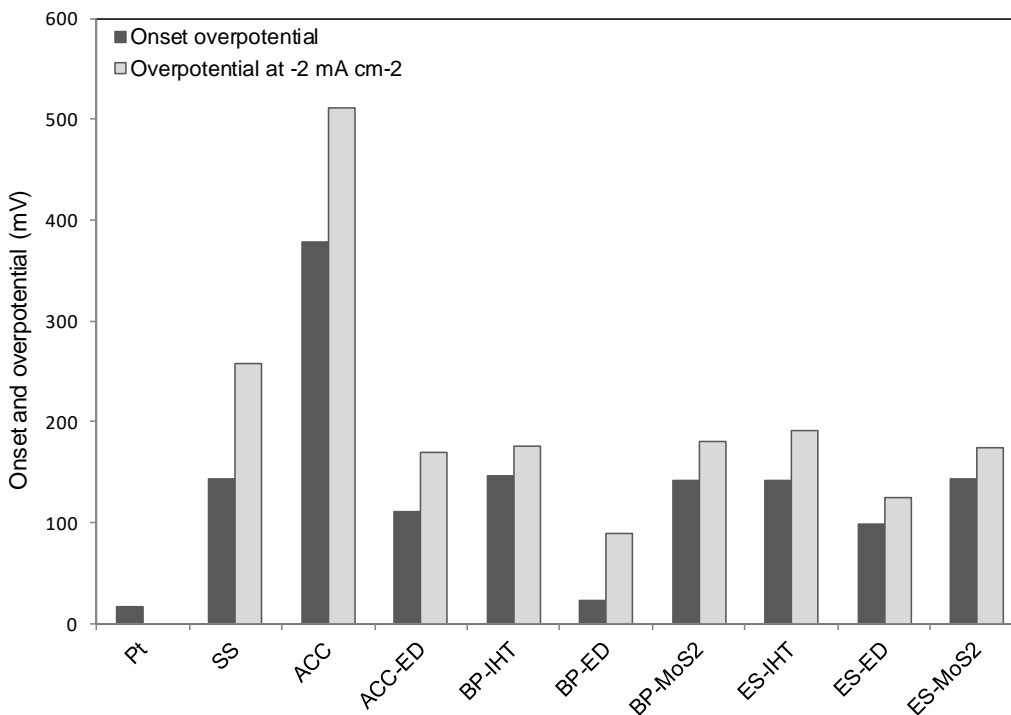
476 Next, the catalytic activity of the MoS_x-coated electrodes towards HER was examined at pH 2.2
477 in 0.5 M phosphate buffer to mimic the pH regime of acidic industrial wastewaters and to
478 identify the best-performing electrodes in this pH regime (Fig. 4, Fig. A.7). In phosphate buffer,
479 the lowest onset overpotential of 20 mV was again observed for the Pt electrode. However, the
480 LSV curve of Pt featured two distinctive waves (Fig. A.5b) that might result from the
481 deprotonation of two different phosphate species (H₃PO₄ and H₂PO₄⁻) at the electrode surface
482 providing protons for H₂ evolution [7]. Due to these current steps, a determination of the
483 overpotential for 2 mA cm⁻² was not possible. The performance of the other reference material,
484 stainless steel, was much poorer in the H₃PO₄ electrolyte compared to the previous runs in
485 H₂SO₄ and resulted in higher overpotentials than for all the other electrodes tested (Fig. 4). Only

486 the untreated ACC support performed even worse and was found to be nearly inactive for HER
487 in phosphate buffer. Among the MoS_x-electrodes, those modified with electrodeposition resulted
488 in the lowest onset overpotentials of 20, 100 and 110 mV with BP-ED, ES-ED and ACC-ED,
489 respectively. In phosphate buffer, the performance of the BP or ES electrodes modified with
490 impregnation followed by heat treatment or constructed by mixing MoS₂ with MWCNTs or to
491 the polymer, respectively, was similar.

492

493 In summary, the seemingly minor shift from the H₂SO₄ (pH 0) to an H₃PO₄ (pH 2.2) electrolyte
494 resulted in major differences of the relative electrocatalytic HER-activities for the different
495 electrode materials: a) in general, overpotentials increase (most likely due to the reduced proton
496 availability), b) the performance of uncoated carbon or steel supports decreases substantially, and
497 – most importantly – c) the differences between the platinum reference and the best MoS_x-
498 electrodes narrows significantly.

499



500

501 **Figure 4** LSV results obtained in 0.5 M phosphate buffer, pH 2.2 (for the abbreviations see
 502 Table 1).

503

504 **3.5 Catalytic activity towards HER in industrial wastewater**

505

506 After establishing some basic HER-performance parameters for the well-defined, acidic H₂SO₄
 507 and H₃PO₄ electrolytes, the electrocatalytic behaviour of the MoS_x-cathodes for HER was
 508 investigated in real industrial wastewater from a plant producing to a large part cellulose acetate.
 509 This wastewater had a pH of 2.4 and contained sulphate and acetate at concentrations of 20-
 510 30 mM and 30-60 mM, respectively. Additionally, the wastewater contains significant amounts
 511 of organic compounds due to the acid treatment of cellulose during the production process.

512

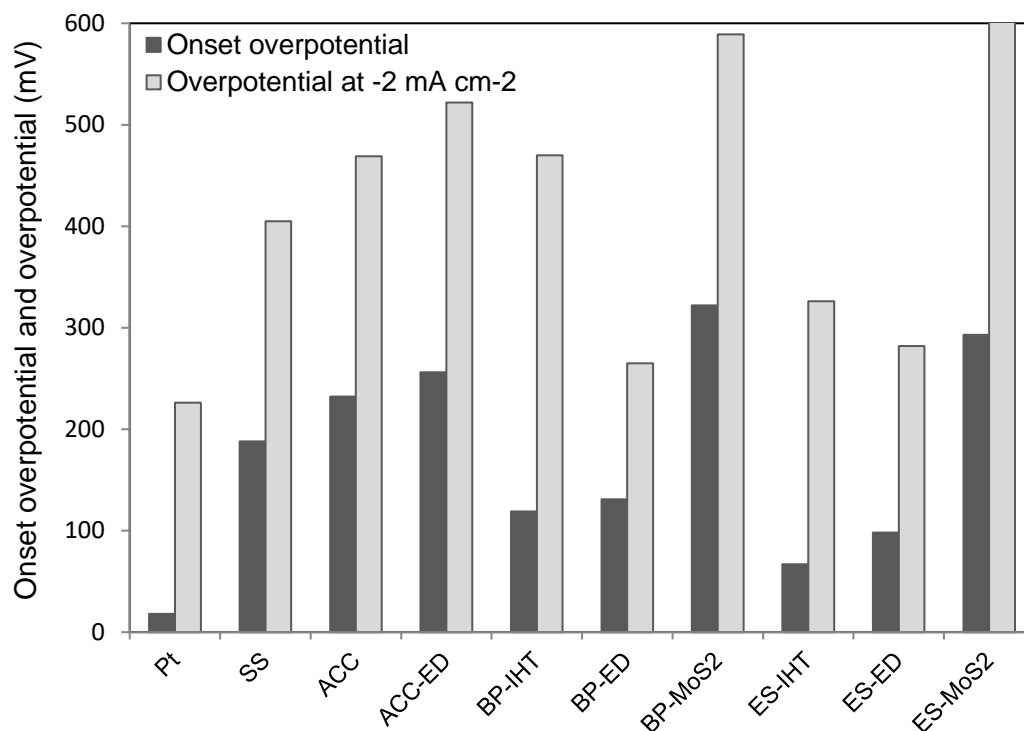
513 The onset overpotentials of the BP and ES electrodes with electrodeposited MoS_x in this
514 wastewater were 130 and 100 mV, respectively (Fig. 5, Fig. A.8) and thus only ~100 mV higher
515 than the onset overpotential of Pt (20 mV) and clearly lower than that of stainless steel
516 (190 mV). Tenca et al. [27] also reported that MoS_2 is a better catalyst for H_2 evolution than
517 stainless steel in industrial or food processing wastewaters at neutral pH. In addition, the
518 overpotentials needed for a reductive current density of 2 mA cm^{-2} for MoS_x -electrodes prepared
519 by electrodeposition were only 40 mV (BP-ED) or 60 mV (ES-ED) higher than that for the Pt
520 electrode. In addition to electrodes modified by electrodeposition, low overpotentials were also
521 detected for BP and ES electrodes where MoS_x was deposited via impregnation followed by heat
522 treatment, which causes the release of ammonia and sulphur from the $(\text{NH}_4)_2\text{MoS}_4$ precursor.
523 This step is often performed at temperatures of 200°C [4], but in the present study milder
524 conditions of only 80°C were sufficient to obtain good catalyst layers. From the different
525 electrodes the ones with MoS_x coating on buckypaper (BP) as support material (i.e. BP- MoS_2 ,
526 BP-IHT and BP-ED) were chosen for the long term experiments after an analysis of the
527 corresponding LSV curves (see Fig. 5 and Fig. A.9).

528

529 When comparing the results for acidic wastewater with the “laboratory electrolytes”, the HER-
530 overpotentials generally increased in the order: sulphuric acid < phosphoric acid < industrial
531 wastewater. The order is, as expected, reversed when comparing current densities: $\text{H}_2\text{SO}_4 >$
532 $\text{H}_3\text{PO}_4 >$ industrial wastewater. Since the results have been corrected for the uncompensated
533 resistance, the variations cannot be explained by differences in conductivity alone. Instead, the
534 catalytic activities of the electrodes for HER in different electrolytes likely vary due to different

535 H₂ evolution reaction mechanisms and/or due to the reduction of compounds other than protons,
536 such as wastewater components.

537



538

539 **Figure 5** LSV results obtained in acidic industrial wastewater, pH 2.4 (for the abbreviations see
540 Table 1).

541

542 **3.6 Long term H₂ evolution from the acidic wastewater electrolyte**

543

544 For long term experiments, Pt and selected buckypaper electrodes were run in
545 chronopotentiometry mode at reductive currents of 3 mA cm⁻² in 0.5 M phosphate buffer or in
546 acidic industrial wastewater for extended time periods (1 - 2.5 weeks) until the cathode potential
547 had stabilised ($\Delta E < 5 \text{ mV d}^{-1}$). The changes of cathode potentials against time for Pt and BP-

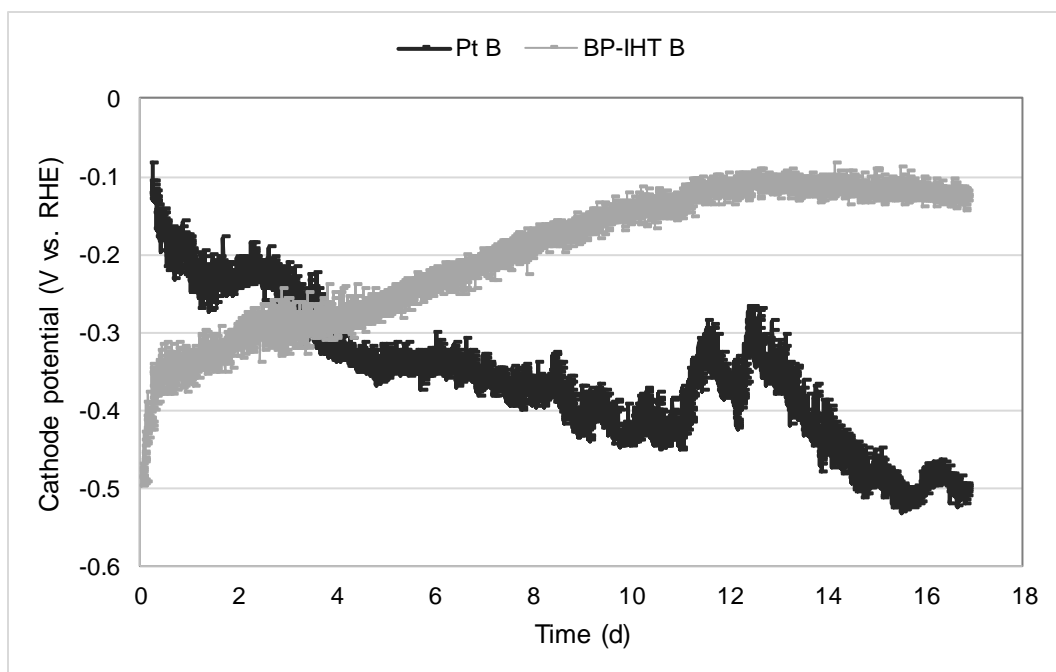
548 IHT electrodes and the overall results from stable operational period are plotted in Fig. 6 and
549 A.10 and in Fig. 7, respectively.

550
551 In phosphate buffer (Fig. 7), the overpotentials for BP-IHT and BP-MoS₂ electrodes surprisingly
552 decreased significantly over time from 220 ± 20 mV to 70 ± 20 mV and from 470 ± 60 mV to
553 260 ± 20 mV, respectively, while for BP-ED electrodes the overpotentials increased from $120 \pm$
554 10 mV to 190 ± 20 mV. The overpotentials of Pt decreased from -55 ± 15 to -81 ± 6 mV during
555 the run. The negative overpotentials for Pt electrodes are likely due to a special H₂ evolution
556 mechanism for Pt electrodes in phosphate buffer. Already the LSVs in phosphate buffer
557 indicated that phosphate species seemed to be able to adsorb onto the platinum surface donating
558 protons for H₂ production (Fig. A.5b). This would result in a lower local pH at the electrode
559 surface and affect the cathode potentials in long term.

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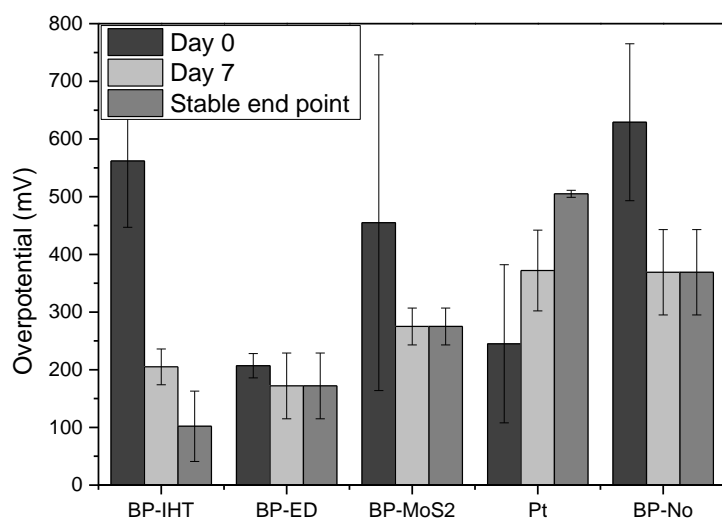
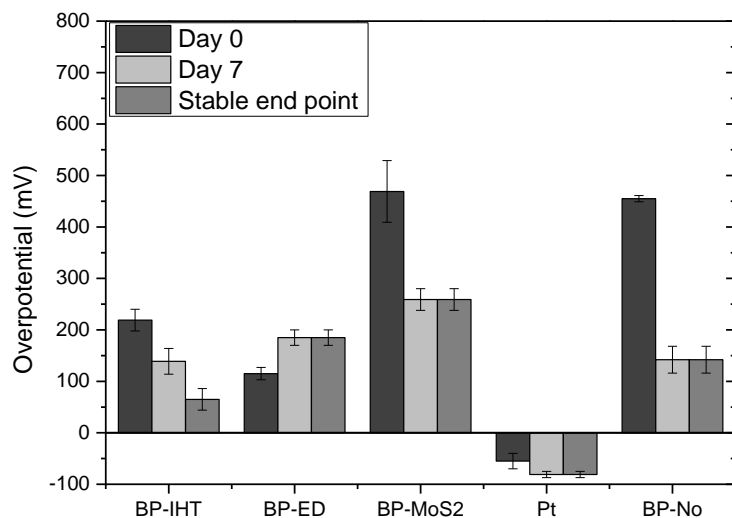
563

564 **Figure 6** Long term experimental run at reductive current of 3 mA cm^{-2} with Pt and BP-IHT
565 electrodes in real acidic industrial wastewater. The figure shows one of the three parallel
566 electrodes and the results have been corrected for uncompensated resistance. For all parallel
567 results, see Fig. A.10.

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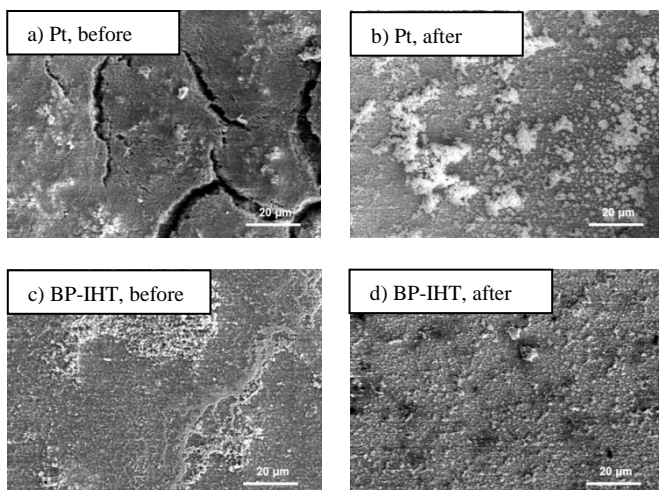
571 **Figure 7** Overpotentials of selected electrodes in long term experiments at reductive current of
 572 3 mA cm^{-2} in 0.5 M phosphate buffer (above) and acidic industrial wastewater (below) at
 573 different time points. “Stable end point” corresponds to the time where the cathode potential had
 574 stabilised ($\Delta E < 5 \text{ mV d}^{-1}$), which took between 7 and 17 days to reach and is indicated
 575 individually above each of the green bars. In each case, standard deviations for three parallel
 576 electrolyses are shown.

577

578 The situation in long term experiments changed dramatically when the same electrochemical
579 runs were carried out in acidic industrial wastewater. Here, the performance of the Pt electrode
580 deteriorated already after some hours as the overpotential increased greatly from 250 ± 140 mV
581 at the start to 510 ± 10 mV after 17 days of continuous electrolysis. Already after 7 days, the Pt
582 cathodes performed worse than all of the investigated MoS_x-based electrodes. After 17 days, the
583 Pt cathodes showed the by far highest overpotential (Fig. 7), even exceeding that of the untreated
584 buckypaper support.

585
586 In order to identify a reason for the performance degradation of the Pt electrodes, SEM images of
587 the electrode surfaces were taken both before and after long term electrolyses. These images
588 clearly show the formation of precipitates on the electrode surfaces, which are more pronounced
589 for Pt than for MoS_x electrodes (Fig. 8). It is known that platinum electrodes can be poisoned by
590 various substances (often sulphur compounds [42,43]). This may explain the increasing
591 overpotentials for the Pt electrodes in our industrial wastewater as it contains high concentrations
592 of sulphate ($2.0 - 2.8$ g L⁻¹) due to the presence of sulphuric acid. At the reducing potentials
593 applied here, SO₄²⁻ could well be reduced to HSO₃⁻ ($E^0 \sim +0.2$ V) and further on to elemental
594 sulphur or sulphides. Alternatively, organic material originating from the cellulose –acetate
595 production process could be deposited and/or adsorbed onto Pt from the high loading of organic
596 matter dissolved in the wastewater.

597



598 **Figure 8** SEM pictures of Pt (a,b) and BP-IHT (c,d) electrodes before use and after long term
 599 run in acidic industrial wastewater.

600

601 In contrast to platinum, an opposite trend was found for the MoS_x electrodes, which substantially
 602 improved their performance over time. For example, the overpotential for BP-IHT electrodes
 603 decreased by nearly half a volt from 560 ± 120 mV to 100 ± 60 mV during 17 days of continuous
 604 electrolysis. A possible explanation could be the slow formation of very active, nano-crystalline
 605 MoS_2 particles during catalysis. This has recently been reported on the basis of results from high-
 606 resolution electron microscopy [44], but a clarification of this issue would require a thorough
 607 investigation. Another possibility might be the conversion of MoS_3 to amorphous MoS_2 , the
 608 latter being known as a better catalyst for HER from previous studies [29]. The conversion of
 609 MoS_3 to MoS_2 can occur, for example, by the reduction of sulphur species from the -I to the -II
 610 oxidation state, which formally results in the conversion of $\text{Mo}^{\text{IV}}(\text{S}^{2-})_1(\text{S}_2^{2-})_1$ to $\text{Mo}^{\text{IV}}(\text{S}^{2-})_2$
 611 accompanied by a release of sulphide, e.g. as H_2S . Due to the deposition of various compounds
 612 from the wastewater on the electrode surface, this could not be proven here by probing the S :
 613 Mo-ratio after the electrolyses by EDX but is definitely an interesting issue for future

614 investigations. The overpotentials of untreated buckypapers also decreased during long term
615 electrolyses both in phosphate buffer and industrial wastewater (Fig. 7). However, the increase
616 was not as large as for MoS_x-coated buckypapers indicating that the improved performance of
617 the MoS_x electrodes was not due to wastewater constituents deposited on the electrodes.

618
619 Overall, these results clearly show that MoS_x electrodes can sustain their high catalytic activity
620 towards HER in real acidic industrial wastewater and that they actually outperform platinum
621 electrodes for H₂ evolution during long term operation. Reasons for this observation might be
622 surface coatings or changes of the MoS_x materials as discussed above. Additionally, it is also
623 possible that this is related to different mechanisms for H₂ formation on Pt versus MoS_x surfaces:
624 for Pt, it has been established that H₂ evolves from hydrogen atoms directly bound to Pt metal
625 centres, while for MoS_x electrodes H₂ formation is thought to occur on the sulphur edges of the
626 MoS_x particles [37]. The mechanistic details of HER on MoS_x are currently much debated in the
627 literature and numerous possible H₂ formation routes are discussed, which e.g. might include
628 Mo^{III}, Mo^{IV} or Mo^V ions, H-H bond formation from SH-groups or protonation steps involving
629 Mo-coordinated hydrides [45–50]. In consequence, it seems possible that catalyst inhibitors
630 present in the wastewater (sulphur species, organic matter, etc.) might affect the platinum surface
631 in a very different way than the active sites of MoS_x, which would explain the dramatically
632 different long term stability of HER-electrocatalysts shown in Fig. 6 and 7.

633 Overall, the results of this study indicate the promising potential of MoS_x electrodes for HER in
634 real industrial wastewaters due to 1) lower overpotentials than Pt during long-term operation in
635 practical conditions, and 2) offering a lower-cost solution for Pt.

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3.6 Hydrogen production

The Faradaic yield for hydrogen production was determined for the two best performing MoS_x-electrodes, BP-ED and BP-IHT, in two chamber electrolysis cells employing Pt mesh as counter electrodes in the anode chambers. Acidic industrial wastewater was continuously pumped through the chamber with a hydraulic retention time of 11.6 h through the chambers to keep the pH stable. The cathodic pH remained under 4.1 during the whole experimental run. The results of headspace gas chromatography measurements showed that H₂ is indeed the main product at the cathode sides with Faradaic H₂ yields (i.e. cathodic hydrogen recoveries) of 90±12% and 91±15% for BP-IHT and BP-ED electrodes, respectively. These hydrogen recoveries are much higher than previously reported for wastewater-fed MEC (16%) [8], but might decrease when the cathodes are combined with biological anodes. H₂ production rates were 0.26 ± 0.03 and 0.39 ± 0.05 m³ m⁻³ d⁻¹ for BP-IHT and BP-ED electrodes with current densities of 1.1 ± 0.1 and 1.7±0.1 mA cm⁻², respectively. Similar H₂ production rates were reported for Pt (0.58 m³ m⁻³ d) and MoS₂ (0.46 m³ m⁻³ d) cathodes in MECs fed with near-neutral industrial wastewaters [27]. On the other hand, the reported rates for stainless steel electrodes in MECs fed with domestic wastewater were much lower (0.007-0.015 m³ m⁻³ d) [8,9].

659 In this study, the MoS_x electrodes prepared with electrodeposition resulted in higher H₂
660 production rates and cathodic H₂ recoveries than the MoS_x electrodes prepared via impregnation
661 followed by heat treatment. However, the H₂ produced was only collected for the first 7 days.
662 The results of the long term experiment (Fig. 7) indicate that the performance of the BP-IHT
663 would still increase in time. Thus, H₂ production with these electrodes should be studied for even
664 longer time in the future. However, this was beyond the scope of this investigation. In this study,
665 H₂ evolution with MoS_x cathodes was examined with abiotic anodes to better evaluate how the
666 use of acidic industrial wastewater as an electrolyte affects H₂ production. In the future, MoS_x
667 cathodes operated in acidic wastewater will be combined with biological anodes where acetate
668 from the same wastewater is oxidised at neutral pH (also available at the industrial site). Having
669 a pH difference between the anode and cathode should not hinder H₂ evolution in MECs. In fact,
670 Ruiz et al. [2] tested different cathodic pH values with synthetic media and reported the highest
671 H₂ production rates for anodic and cathodic pH values of 7.0 and 2.0, respectively.

672

673 4. Conclusions

674

675 We have demonstrated that MoS_x can clearly outperform platinum as catalyst for H₂ evolution in
676 a real acidic industrial wastewater with regard to long term stability. The best performing MoS_x-
677 electrodes also had lower onset overpotential for H₂ evolution in wastewater (100-130 mV) than
678 stainless steel electrodes (190 mV) which are often used for H₂ production in microbial
679 electrolysis cells at present. In addition, the catalytic activity of the MoS_x electrodes for HER
680 increased in industrial wastewater over time, reaching overpotentials as low as 100 mV (after 17
681 days) at a current density of 3 mA cm⁻². The opposite was observed for Pt electrodes where the
682 overpotentials needed to sustain this current density had to be increased from 250 to 510 mV
683 over the same time period. The reasons for the decreasing overpotentials of the MoS_x electrodes
684 need further investigation but might be related to changes in the MoS_x structure and/or the
685 sulphide-to-disulphide (S²⁻ : S₂²⁻) ratio on the surface of the catalyst. To be able to optimize
686 catalytic performance in a rational way, more mechanistic studies are needed in the future [45–
687 50]. We also found that the MoS_x deposition method as well as the carbon support greatly
688 affected the catalytic activities of the electrodes for HER. Electrodeposition and impregnation
689 followed by heat treatment resulted in lowest overpotentials of the MoS_x electrodes and MoS_x
690 electrodes prepared with these methods resulted in high cathodic H₂ recoveries (around 90%)
691 and H₂ production rates (0.26-0.39 m³ m⁻³ d). Thus, MoS_x-carbon electrodes offer a viable option
692 for hydrogen evolution in MECs under practical conditions.

693 The possible range of practical application in the field of energy conversion is manifold. As
694 mentioned before, the MoS_x-carbon electrodes are suitable for the application as H₂ evolution
695 cathodes in microbial electrolysis cells treating complex acidic wastewaters, in which platinum

696 cathodes suffer from severe poisoning. Using the presented MoS_x-cathodes would enable to
697 reduce the total cell voltage of an MEC needed for long-term operation to only 0.5 V, as
698 compared to the approx. 0.9 V required when using the poisoning prone platinum cathodes for
699 longer than a couple of days. This corresponds to an by approx. 40% lower electricity demand
700 for hydrogen production (calculation based on a reasonable operation potential of ~ + 0.4 V vs.
701 RHE for a microbial anode [51] and the cathode operating potentials of ~ - 0.1 V and - 0.5 V vs.
702 RHE after 17 days of operation as displayed in Fig. 6). Furthermore, the new cathodes could also
703 be applied in the emerging field of microbial electrosynthesis [52], were electrical energy is
704 converted into valuable products by hydrogenotrophic microorganisms. Examples are the
705 methanization of CO₂ from biogas [53], and the production of commodities such as acetate or the
706 bioplastic poly-β-hydroxybutyrate from HCO₃⁻ as carbon source [52]. These energy conversion
707 processes require hydrogen evolution cathodes that are poisoning-resistant in a complex
708 microbial environment. Considering the results of this study, MoS_x might be a very suitable
709 catalyst material for this task. In the light of the energy transition towards an electricity powered
710 world [54], it is expected that microbial electrosynthesis will gain significant future relevance to
711 convert electricity into valuable products [52]. We strongly believe that the bioelectrochemical
712 route can favourably complement the more conventional “Power-to-X”-technologies [55], that
713 are increasingly being researched.

714 From an application point of view it is particularly important, that the presented electrode
715 fabrication routes can, in principle, be realized at low cost also at larger scale. For example, the
716 carbon-nanotube-based support of the best-performing electrode is fabricated as buckypaper by a
717 simple filtration method similar to paper fabrication. Considering a CNT-loading of 12 g m⁻² and
718 taking into account the current price of 199 US\$ per kg of industrial grade multiwalled-CNTs

719 (20-40 nm diameter, <https://www.cheaptubes.com>, accessed on 14.11.2016), its material cost
720 amounts to only 2.38 US\$ m⁻². Similarly, the investigated MoS_x-deposition methods (drop
721 coating, impregnation and heat treatment, electrodeposition) are feasible at larger scales. With a
722 MoS_x loading of 20 mmol m⁻² and costs of ~1.5 US\$·(mol Mo)⁻¹ (<http://www.lme.com>, accessed
723 on 26.01.2016) the molybdenum costs amount to only 0.03 US\$ per m² electrode, resulting in a
724 total material cost of the electrode of only ~2.40 US\$ per m². For comparison, just the Pt cost of
725 a 1 m² electrode is already 58 times higher, amounting to 141 US\$. (5 mg m⁻² Pt loading; ~5500
726 US\$·(mol Pt)⁻¹, <http://www.lme.com>, accessed on 26.01.2016). Considering that the MoS_x-
727 cathodes exhibit a four-fold lower overpotential for HER, the results of this study clearly
728 demonstrate the tremendous potential of MoS_x electrodes as efficient and affordable cathodes for
729 HER in real industrial wastewaters.

730

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